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# **Sodium Oxide-Phosphorus( V) Oxide-Water Phase Diagram near 300** *O* **C: Equilibrium Solid Phases**

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The products of evaporation at 290 °C of sodium phosphate solutions with Na/P ratios ranging from 1.6 to 9 have been recovered under controlled humidities for examination by X-ray powder diffraction. The equilibrium solid phases inferred from these experiments are combined with previously reported solubility data to derive a detailed description of the Na<sub>2</sub>O-P<sub>2</sub>O<sub>5</sub>-H<sub>2</sub>O ternary phase diagram near 300 °C. X-ray diffraction data are reported for the following previously unreported or incompletely characterized phases: Na<sub>3</sub>PO<sub>4</sub>.NaOH, Na<sub>3-x</sub>H<sub>x</sub>PO<sub>4</sub>.xH<sub>2</sub>O (three forms), Na<sub>2.5</sub>H<sub>0.5</sub>PO<sub>4</sub>.zH<sub>2</sub>O, and a previously unreported polymorph of Na<sub>2</sub>HPO<sub>4</sub>-NaH<sub>2</sub>PO<sub>4</sub>. The structural relationship between Na<sub>3-x</sub>H<sub>2</sub>PO<sub>4</sub>.xH<sub>2</sub>O and the high-temperature modification of Na<sub>3</sub>PO<sub>4</sub> is discussed.

#### **Introduction**

A detailed knowledge of the compositions of saturated sodium phosphate solutions near 300 °C is important for optimizing chemistry control in steam-generating boilers.'-8 Previous fundamental studies of the  $Na_2O-P_2O_5-H_2O$  system have been summarized by Broadbent et al.<sup>1</sup> Although three solubility studies of this system in the temperature range 250-365 °C have been reported,<sup>1-3</sup> the nature of equilibrium solid phases is still incompletely understood. A phase of variable composition, approximating to  $11Na_2O\cdot 4P_2O_S\cdot 3H_2O$ (or  $\text{Na}_{2.75}\text{H}_{0.25}\text{PO}_4 \cdot 0.25\text{H}_2\text{O}$ ), is the equilibrium solid over a wide range of saturated solution compositions<sup>1-3</sup> (molar ratios in the range  $\sim$  2.2  $\leq$  Na/P  $\lt$  at least 9). Near Na/P = 2.0, Na<sub>2</sub>HPO<sub>4</sub> and Na<sub>4</sub>P<sub>2</sub>O<sub>7</sub> are equilibrium phases at 250 and 300  $\textdegree C$ , respectively.<sup>1</sup> The system near Na/P = 2.0 is further complicated by liquid immiscibility.'

Although diffraction studies of a very large number of sodium phosphates have been reported, $9$  crystallographic information pertinent to this study is restricted to powder data for Na<sub>3</sub>PO<sub>4</sub>,<sup>9-13</sup> Na<sub>2</sub>HPO<sub>4</sub>,<sup>9</sup> Na<sub>2</sub>HPO<sub>4</sub></sub>.NaH<sub>2</sub>PO<sub>4</sub>,<sup>9</sup> Na<sub>4</sub>P<sub>2</sub>O<sub>7</sub>,<sup>9</sup>  $Na_3PO_4.0.5H_2O$ ,<sup>10</sup> and  $11Na_2O.4P_2O_5.3H_2O.1$  Anhydrous trisodium phosphate undergoes a reversible, probably nonreconstructive transformation from a tetragonal  $(\alpha)$  to a cubic *("7'')* structure near **300 0C.1D-12** The latter is believed to be isostructural with  $Na<sub>2</sub>CaSiO<sub>4</sub>$ , but precise atomic positional parameters have not been determined.<sup>10</sup> Kizilyalli and Welch<sup>13</sup> reported two orthorhombic forms of Na<sub>3</sub>PO<sub>4</sub>, but these have not been observed by other workers.<sup>10-12</sup> Mair<sup>10</sup> and Hooper et al.<sup>14</sup> have shown that cubic  $Na<sub>3</sub>PO<sub>4</sub>$  has a considerable capacity for solid solution formation by substitution of either anions or cations. These solid solutions tend to retain the cubic structure at room temperature. Polymorphism has not been reported for any other anhydrous sodium orthophosphates.

The reported solubility studies<sup>1-3</sup> are in good general agreement, but practical limitations in the accuracy of such studies restrict the information which they can provide about equilibrium solid phases. We therefore adopted an alternative, complementary approach in which sodium phosphate solutions in the composition range  $1.6 \leq \text{Na}/\text{P} \leq 9.3$ , including values approach in which solutions we complementary approach in which sodium phosphate solutions near known congruent and invariant points, were evaporated

to dryness near 300 °C, and the solid products were analyzed by X-ray powder diffraction. The equilibrium solid phases inferred from these "drying up" experiments could then be used to confirm or expand the details of the solubility-derived phase diagram.

#### **Experimental Section**

Careful humidity control during recovery of the precipitated phases was essential, in order to avoid the formation of polyphosphate or hydrated phases. The experiments were conducted in two independently controlled Parr Series 4601, 1 dm', 316 stainless steel autoclaves shown in Figure 1, each with its own Parr 4913 heater assembly, a Love 4821 temperature controller, and a 0-3000 psi pressure gage. The autoclaves were connected by stainless steel tubing wrapped in heating tape to prevent condensation and could be isolated from one another by two Whitey SS-3VS4 high-temperature valves. The sample autoclave was loaded with about  $300 \text{ cm}^3$  of sodium phosphate solution whose composition was calculated to be about 20% below saturation at 300  $^{\circ}$ C.<sup>2</sup> The solutions were poured both inside and outside the Zircaloy 2 can which was then covered by a loose top to prevent water which had condensed on the autoclave head from entering the can during cool-down. With the more concentrated solutions of lower Na/P ratios, water, rather than solution, was poured outside the can, to reduce corrosion of the autoclave. The autoclave was left overnight at 300 °C before commencing the evaporation. The temperature was measured and controlled by a stainless-steel-jacketed, iron-constantan thermocouple which passed through a Conex MIC-062-AL thermocouple fitting in the autoclave head into a well at the bottom of the can. Solutions were evaporated to dryness by slowly releasing water vapor through the condenser. The autoclave temperature fell during the evaporation to  $290 \pm 5$  °C and was consistent with the observed water vapor pressure as long as liquid solution was present. A brief overshoot (<15 min) of up to  $50^{\circ}$ C occurred just as the abrupt fall in the vapor pressure indicated that the final traces of liquid had evaporated. This did not appear to cause appreciable decomposition of the samples. In general, the material which precipitated inside and outside the Zircaloy can had similar X-ray diffraction patterns, but the latter was visibly colored and more poorly crystalline.

The water autoclave contained doubly distilled water and was used to inhibit the decomposition of sodium hydrogen orthophosphates to pyrophosphates at Na/P ratios below 3.00 by maintaining a given partial pressure of water vapor in the sample autoclave. In practice, the second autoclave was maintained at a temperature about 15 °C



**Figure 1.** Autoclave assembly.

below the first during the evaporation and cooling steps. Water vapor from it was vented into the sample autoclave just as final evaporation occurred. As cooling proceeded below 150 "C, the sample autoclave was again isolated and was flushed overnight with dry nitrogen before removing the precipitated solids in a dry nitrogen atmosphere.

Solutions were prepared from Baker Analyzed reagent grade NaH<sub>2</sub>PO<sub>4</sub>.H<sub>2</sub>O and Na<sub>2</sub>HPO<sub>4</sub> and from Fisher Certified standard NaOH solutions, with use of doubly distilled water. The NaOH standardization was checked by a potentiometric titration with HCl, and the phosphates' water of hydration was determined by drying at 120 °C. Carbon dioxide free water was used to prepare solutions with Na/P ratios greater than **2.00.** 

Powder X-ray diffraction data for isolated solids were obtained at temperatures from 20 to 285 °C with a Nonius Model FR553L Guinier-Simon variable-temperature focussing camera with quartz-monochromated Cu *Ka,* radiation, a Nonius Model **FR508V**  Guinier film measuring device, and a Nonius Model **I1** microdensitometer. Na<sub>2</sub>HPO<sub>4</sub> was used as a calibration standard for the  $d$ spacings.<sup>9</sup> Some room-temperature diffraction studies were repeated, with larger samples, in a Philips Model PW1050 powder diffractometer to ensure that these results were representative of the bulk material. Relative amounts of Na<sub>2</sub>HPO<sub>4</sub> and Na<sub>4</sub>P<sub>2</sub>O<sub>7</sub> in samples recovered from solutions with  $\text{Na}/\text{P} = 2.00$  were estimated from relative major X-ray diffraction peak intensities by using standard mixtures.

The water content of some products was determined by thermogravimetric analysis with a Perkin-Elmer TGS-1 thermobalance purged with a dry nitrogen flow of about **30** cm3 min-' and a heating rate of 20 °C min<sup>-1</sup>. The relative amounts of orthophosphate and polyphosphates in material recovered at  $Na/P = 1.6$  were determined by gradient elution chromatography.<sup>15</sup>

The following methods were used in attempts to synthesize some of the solids recovered from solution. Tetragonal, anhydrous  $Na_3PO_4$ was prepared by fusing Na<sub>2</sub>HPO<sub>4</sub> and Na<sub>2</sub>CO<sub>3</sub> in a Pt boat at 600 **OC, in accordance with eq 1.** This reaction was also performed in  $2Na_2HPO_4 + Na_2CO_3 \rightarrow 2Na_3PO_4 + H_2O + CO_2$  (1)

$$
2Na_2HPO_4 + Na_2CO_3 \rightarrow 2Na_3PO_4 + H_2O + CO_2 \qquad (1)
$$

the presence of NaOH, NaF, and NaCl to test the possibility of adduct formation between  $Na<sub>3</sub>PO<sub>4</sub>$  and those salts. The solid-state reaction between NaOH and tetragonal Na<sub>3</sub>PO<sub>4</sub> was monitored between 20 and 285 °C by using the X-ray techniques described below. The double salt  $Na<sub>2</sub>HPO<sub>4</sub>·NaH<sub>2</sub>PO<sub>4</sub>$  was prepared by evaporation of a solution with  $Na/P = 1.50$  in a laboratory oven at 110 °C.

#### **Results**

The solid phases recovered from solutions of various compositions are listed in Table I and powder X-ray diffraction data for the previously unreported or partially characterized phases are given in Tables **11-IV** and in Tables **V** and **VI** in the supplementary material. The sodium to phosphate mole ratios, Na/P, which are used in Table **I** and the following discussion, refer to the initial compositions of the solutions.

At Na/P = 9.3 the major phases were  $\alpha$ -NaOH and NaOH $\cdot$ H<sub>2</sub>O. A minor phase at Na/P = 9.3 (designated I) was identical with the major product at Na/P = 4.0 and 4.3. We recovered  $\alpha$ -Na<sub>3</sub>PO<sub>4</sub> as a major product at Na/P = 3.15, and it was also present at  $Na/P = 3.00$ . The previously unreported phase, **I,** was also observed in the reaction of solid NaOH with  $\alpha$ -Na<sub>3</sub>PO<sub>4</sub> at temperatures above 90 °C. We conclude that it is a double salt,  $Na<sub>3</sub>PO<sub>4</sub>·xNaOH$ , with x close

**Table 1.** Precipitate Compositions  $(r = 290 \pm 5 \degree \text{C} \text{ Unless}$ Specified Otherwise<sup>d</sup>)

	major	minor	$P_{\rm H_2O}^{\qquad b}$ /
Na/P <sup>a</sup>	products	products	MP <sub>a</sub>
9.30	NaOH, NaOH·H, O		
4.30		NaOH·H, O	
4.00		$\alpha$ -Na <sub>2</sub> PO <sub>2</sub>	
3.15	$\alpha$ -Na <sub>2</sub> PO <sub>4</sub>	1	
3.00	HВ	$\alpha$ -Na <sub>2</sub> PO <sub>4</sub>	
2.83	ПA		7
2.75	ПA	ш	7
2.65	IIA, III	НC	$\overline{7}$
2.50	ш	$Na, HPOa$ ,	$\overline{7}$
		unident <sup>c</sup>	
2.40	III, Na,HPO,	$NaaP2O7$	7
2.15	Na <sub>2</sub> HPO <sub>4</sub>	ш	7
2.00	$Na$ , $HPO$ ,	$Na4P2O2$ $(\sim\!20\%)$	7
2.00 <sup>d</sup>	$Na4P2O2$	Na <sub>2</sub> HPO <sub>4</sub> $(\sim 20\%)$	
1.80	Na, HPO <sub>4</sub>	unident	
1.60	Na <sub>2</sub> HPO <sub>4</sub> ·NaH <sub>2</sub> PO <sub>4</sub>	unident	7
1.60	$\text{Na}_4\text{P}_2\text{O}_7$ , amorphous orthophosphate		4

a Na/P mole ratio in initial, unsaturated solutions. b Minimum water partial pressure during evaporation at temperature *I* when the second autoclave was used to control humidity.  $c$  unident = unidentified crystalline products.  $d \text{Na}_4P_2O_7$  formed initially by evaporation at **330** 'C, to test for formation of Na,HPO, during recovery.

Table II. X-ray Diffraction Data for Phase I, Na<sub>3</sub>PO<sub>4</sub>.NaOH at **22** 

hkl	$d_{\mathrm{calcd}}$ Åδ	$d_{\textbf{obsd}}/$ A	$I/I_1^{\mathbf{c}}$	hkl	$d_{\text{calcd}}/$ Å	$d_{\text{obsd}}/$ Å	$I/I$ ,	
101	5.954	5.952	17	031	2.302	2.303	15	
011	5.354	5.355	8	203	2.287	2.287	46	
111	4.592	4.588	42	312	2.266	2.266	7	
200	4.464	4.464	7	400	2.232	2.233	40	
002	3.994	3.990	22	321	2.206	2.206	45	
201	3.897	3.897	32	230	2.117	2.118	$\overline{\mathbf{4}}$	
210	3.796	3.795	46	123	2.083	2.084	32	
102	3.646	3.643	7	132	2.007	2.008	18	
020	3.606	3.603	6	322	1.990	1.992	35	
211	3.429	3.427	$\leq$	303	1.985	1.986	10	
112	3.254	3.255	42	104	1.949	1.950	71	
121	3.085	3.083	26	313	1.913	1.913	6	
202	2.977	2.979	17	232	1.870	1.871	17	
300	2.976			330	1.870			
220	2.805	2.805	11	421	1.847	1.846	$<$ 2	
301	2.789	2.789	13	204	1.823	1.824	6	
212	2.752	2.754	5	040	1.803	1.804	47	
221	2.647	2.647	59	500	1.786	1.786	20	
311	2.601	2.601	5	033	1.785			
122	2.564	2.564	100	140	1.768	1.769	20	
103	2.552	2.551	15	214	1.768			
013	2.498	2.500	11	501	1.743	1.742	8	
113	2.406	2.408	$<$ 2	124	1.715	1.716	16	
				422	1.714			

 $a$  Weak lines due to traces of phase IIA and  $\alpha$ -Na, PO, were Orthorhombic; *a* = omitted; 34 further lines to  $d_{\text{obsd}} = 1.280$  Å could not be uniquely indexed and are also omitted. **8.929 (3),** *b* = **7.213 (2),** *c* = **7.989 (2) A.**  Determined photometrically.

to 1.0, since it was the major product (ca. 90%) of both solid-state reactions and autoclave experiments at  $Na/P = 4.0$ . 1.0, since it was the major product (ca. 90%) of both<br>lid-state reactions and autoclave experiments at Na/P = 4.0.<br>The situation in the region of composition 2.00 < Na/P  $\leq$ <br>200 is very complex and two phases one of whi

3.00 is very complex, and two phases, one of which displays variable composition, have been observed in addition to  $Na<sub>3</sub>HPO<sub>4</sub>$  (at  $Na/P \le 2.50$ ) and  $\alpha$ -Na<sub>3</sub>PO<sub>4</sub> (at Na/P = 3.00). The phase of variable composition (here designated **11)** gives diffraction patterns similar to that reported by Broadbent et al.' for the material which they isolated from solutions with

Table III. X-ray Diffraction Data for Phase IIA,  $Na_{3-x}H_xPO_4 \cdot xH_2O$ , and Cubic  $Na_3PO_4{}^a$ 

phase IIA $(22^{\circ}C)^b$			phase IIA (285 $^{\circ}$ C) <sup>b</sup>			cubic $Na3PO4$ (25 °C)				
$hkl$	$d_{\rm{calcd}}/\rm{A}^c$	$d_{\text{obsd}}/\text{A}$	$I/I_1^d$	hkl	$d_{\rm{calcd}}/A^e$	$d_{\text{obsd}}/A$	$\boldsymbol{I}$	hkl	$d_{\rm{calcd}}/A^f$	I/I
$11\overline{1}$ $1\overline{1}1$ ī11	4.368 4.281 4.232	4.372 4.278	$40-$ 30	$\frac{111}{111}$	4.375 4.300	4.371 4.296	$\binom{m}{s}$	111	4.279	70
111 200	4.2345 3.760	4.233 3.763	لـ 70 $20$ ך	200	3.740	3.739	m, b	200	3.705	10
020 002 220	3.687 3.669 2.653	3.689 3.670 2.653 <sub>1</sub>	15 <sub>1</sub> $15 \; J$							
$20\overline{2}$ $0\overline{2}2$ $2\overline{2}0$	2.6451 2.642 <sub>5</sub> 2.613	1 2.641 2.614 <sub>1</sub>	$100 -$	220 $\overline{2}20$	2.662 2.628	2.662 2.627	$vs, b \uparrow$ <b>VS</b>	220	2.620	100
202 022	2.607 2.561 $13\overline{1}$ <sup>g</sup> 2.258	$2.604\,\mathrm{K}$ 2.560 2.257	80 40 157							
	$3\overline{1}1^g$ 2.246 $0\overline{3}2^g$ 2.0721 230 <sup>g</sup> 2.0725	2.246 2.072	5J 8 <sup>7</sup>		(unobsd) (unobsd)			311 320	2.234 2.055	0 $\bf{0}$
400	$\overline{2}03g$ 2.064 1.880	2.063 1.881	8 <sub>1</sub> 20 <sup>7</sup>							
040 004	1.843 1.834 1.548	1.843 1.835 1.549	15 15J $7\overline{7}$	400	1.870	1.871	s, b	400	1.853	20
$422$ $242$ $422$ $422$ $422$ $2422$ $2422$ $224$	1.537 $1.532\,$ 1.517	1.535	11							
	1.516 1.515 1.514	1.516	20	422 $4\overline{2}2$	1.543 1.524	1,543 1.524	m m, b	422	1.513	20
242	1.510 1,497)	1.509 1.495	$\overline{7}$ 10	422	1.517	1.517	m			
224	$\overline{2}24$ 1.4935 242 1.489 1.486	1.489 1.485	$\begin{matrix} 5 \\ 5 \end{matrix}$							

<sup>a</sup> Square brackets indicate reflections which are degenerate in the higher symmetry phases; m = medium, s = strong, v = very, b = broad.<br><sup>b</sup> These diffraction patterns also contained a large number of weak lines, most of be due to minor phases, but some may be due to a superlattice structure. With the exception of one line at  $d_{\text{obsd}} = 3.110 \text{ A } (I/I_1 = 9)$ , all unindexed lines had  $I/I_1 \le 5$ . <sup>C</sup> Triclinic;  $a = 7.522$  (3),  $b = 7.378$  (3) mined photometrically; the components of some barely resolved doublets were estimated visually on the basis of the integrated photometric<br>intensity for the two lines. <sup>e</sup> Rhombohedral;  $a = 7.482$  (2) A;  $\alpha = 89^{\circ}$  13 (2

initial compositions in the range  $2.14 \leq Na/P \leq 9.05$  and reported as  $11Na<sub>2</sub>O<sub>3</sub> + 3H<sub>2</sub>O<sub>4</sub>$ . A reproducible, sharp diffraction pattern was obtained from material recovered from solutions of initial composition  $Na/P = 2.83$ , close to a known congruent point. $1-3$  The same phase was the major component at  $\text{Na/P} = 2.75$  and was also present at  $\text{Na/P} = 2.65$ ; it is denoted IIA (Table 111). A similar but not identical phase, IIB (Table IV), was the major product recovered at Na/P = 3.00. The diffraction patterns of IIB, and of some samples of IIA, displayed some diffuse scattering in the vicinity of major peaks, suggesting compositional heterogeneity. A third variant of phase 11, denoted IIC (Table IV), was a minor product at  $Na/P = 2.65$ .

A minor component at Na/P = **2.75** was recovered in larger quantities at  $\text{Na}/\text{P} = 2.65$  and was also obtained at  $\text{Na}/\text{P} =$ **2.50, 2.40,** and **2.15** (together with increasing quantities of  $Na<sub>2</sub>HPO<sub>4</sub>$ ). This material is referred to as phase III. Although phase I11 has not been isolated, and so could not be analyzed, it was recovered as the major component of a mixture with  $Na<sub>2</sub>HPO<sub>4</sub>$  at  $Na/P = 2.4$ , so that its diffraction pattern could be distinguished (Table V in supplementary material). No variations in this diffraction pattern were observed with samples recovered from solutions with compositions in the range  $2.15 \leq \text{Na/P} \leq 2.65$ , indicating a narrow range of stoichiometry for III and a miscibility gap between I1 and 111. The relative quantities recovered from different experiments indicate that I11 has a Na/P04 ratio close to **2.5,**  and we tentatively assign to I11 the formulation  $Na<sub>2.5</sub>H<sub>0.5</sub>PO<sub>4</sub>·zH<sub>2</sub>O$ , with z probably having a value between 0 and **0.5.** 

The only phases observed at  $Na/P = 2.0$  were  $Na<sub>2</sub>HPO<sub>4</sub>$ and  $Na<sub>4</sub>P<sub>2</sub>O<sub>7</sub>$ . The relative quantities of these two phases were very sensitive to experimental conditions, as discussed in the following section. At  $Na/P = 1.6$ , the major phase was  $Na<sub>2</sub>HPO<sub>4</sub>·NaH<sub>2</sub>PO<sub>4</sub><sup>9,16,17</sup>$  and an unidentified minor phase was also observed. At  $\text{Na}/\text{P} = 1.8$ , we obtained a mixture of  $\text{Na}_2$ HPO<sub>4</sub>, the same minor phase that was observed at  $\text{Na}/\text{P}$ = 1.6, and at least one other unidentified phase. Since suppression of polyphosphate formation becomes increasingly difficult with decreasing Na/P ratio (see Appendix in supplementary material), these unidentified phases may have been products of decomposition of equilibrium phases.

X-ray diffraction of material recovered from a solution with  $Na/P = 1.6$  under a lower partial pressure of water from the water autoclave (4MPa at 290 °C) revealed only Na<sub>4</sub>P<sub>2</sub>O<sub>7</sub>. However, gradient elution chromatography<sup>15</sup> of this material revealed considerable amounts of orthophosphate as well as pyrophosphate. The orthophosphate was presumably present as an amorphous solid. This is consistent with the work of Edwards and Herzog,<sup>16</sup> who observed  $Na_4P_2O_7$  and an amorphous orthophosphate as intermediate products of the dehydration of a sodium orthophosphate mixture with  $Na/P$  $= 1.67$ . Autoclave corrosion became severe at Na/P  $\leq 1.8$ , due to the high solubility and low pH of the solutions, so that measurements below  $\text{Na}/\text{P} = 1.6$  were not practical.

In the course of this work, we attempted to prepare Na<sub>2</sub>HPO<sub>4</sub>.NaH<sub>2</sub>PO<sub>4</sub> by evaporation of a solution with Na/P = 1.5 at 110 °C. This material was found to have a diffraction pattern different from that reported for  $Na<sub>2</sub>HPO<sub>4</sub>~NaH<sub>2</sub>PO<sub>4</sub>$ <sup>9</sup> but to transform to that material without significant loss of

**Table IV.** X-ray Diffraction Data for  $\text{Na}_{3-x}\text{H}_{x}\text{PO}_{4} \cdot x\text{H}_{2}\text{O}$  Phases IIB and IIC<sup>a</sup>

	phase IIB $(22 °C)^b$				phase IIB (285 °C) <sup>c</sup>	phase IIC $(22 °C)^d$		
hkl	$d_{\rm{calcd}}/\rm{A}$	$d_{\textbf{obsd}}/\text{Å}$	$I/I_1^e$	$d_{\rm{calcd}}/\rm{\AA}$	$d_{\rm obsd}/\rm A$	$d_{\rm{calcd}}/\rm{A}$	$d_{\text{obsd}}/A$	$I\!/\!I_1$
$11\overline{1}$	4.330	4.329	45	4.353	4.358			
$\overline{1}\underline{1}\underline{1}$	4.296	4.296	70	4.326	4.324			
$1\overline{1}1$	4.278	4.273 <sub>5</sub>		4.308	4.306			
111	4.229	4.229	35	4.267	4.265	4,308	4.308	S.
200	3.751	3.751	20	3.775	3.775	3.731	3.732	${\bf m}$
020	3.733	3.732	20	3.755	3.757			
002	3.645	3.646	15	3.677	3.678			
$\overline{2}20$	2.650	2.651		2.666	2.666 <sub>1</sub>	2.638	2.637	${\bf S}$
220	2.6421	2.642	100	2.6591	2.658'			
202	2.642 <sub>5</sub>			2.6585				
$0\overline{2}2$	2.627	2.627	40	2.643	2.642			
022	2.589	2.588	65	2.612	2.612			
202	2.587f			2.611				
$\overline{1}31^f$	2.247	2.246	10	2.259	2.258			
$2\overline{2}2^f$	2.139	2.141	10	2.154	2.154			
400	1.876	1.876	15	1.888	1.888	1.865	1.865	m
040	1.866	1.867	20	1.878	1.878			
004	1.823	1.823	20	1.839	1.839			
$\overline{4}\overline{2}2$	1.536	1.536	$\mathbf{2}$	1.544)				
	1.531	1.530	5	$1.541$ }	1.540 vb			
$\frac{1}{242}$ $\frac{242}{242}$	1.530 f			1.539				
	1.523			1.532				
$\frac{242}{224}$ $\frac{224}{222}$	1.518	1.517 b	10	1.529				
	1.518			$1.527\}$	1.528 b			
	1.517			1.528				
422	1.507)			1.520				
242	1.505	1.505 b	5	$1.517$ $>$	1.517 b			
$\overline{2}24$	1.505			1.517				
$2\overline{2}4$	1.499	1.497	$\boldsymbol{2}$	1.511	1.509			
224	1.483	1.482	2	1.498	1.498			

*a* Weak, unindexed lines  $(I/I_1 < 5)$ , some of which may be superlattice reflections, have been omitted. *b* Triclinic;  $a = 7.505$ ,  $b = 7.467$ ,  $c =$ 7.292 A;  $\alpha$  = 90.8,  $\beta$  = 91.1,  $\gamma$  = 90.1°. CTriclinic; a = 7.552, b = 7.511, c = 7.357 A;  $\alpha$  = 90.6,  $\beta$  = 91.0,  $\gamma$  = 90.1°. Cubic; a = 7.461 A. **e** Determined photometrically for 22 "C data; relative intensities at 285 "C are similar. *f* Tentative indexing.

weight at  $\sim$ 120 °C. This transformation was irreversible, and we conclude that the initial product is a metastable polymorph of  $Na<sub>2</sub>HPO<sub>4</sub>·NaH<sub>2</sub>PO<sub>4</sub>$ . Its diffraction pattern is given in Table VI (supplementary material), Its formulation was confirmed by the weight loss on heating to 450 "C, according to eq 2. The identity of the products of reaction 2 was

constrained by X-ray diffraction.

\n
$$
2(Na_2HPO_4 \cdot NaH_2PO_4) \rightarrow Na_5P_3O_{10} + NaPO_3 + 3H_2O_{20}
$$
\n(2)

#### **Structural Inferences**

**Phase I.** The X-ray diffraction pattern for I,  $Na<sub>3</sub>PO<sub>4</sub>$ . NaOH, in Table I1 was indexed on the basis of an orthorhombic unit cell:  $a = 8.929$  (3),  $b = 7.213$  (2),  $c = 7.982$ (2)  $\hat{A}$ . The unit cell volume, 514.5  $\hat{A}^3$ , is close to 530.3  $\hat{A}^3$ , the value calculated from the volumes of  $\alpha$ -Na<sub>3</sub>PO<sub>4</sub> and  $\alpha$ -NaOH, assuming four formula units of Na<sub>3</sub>PO<sub>4</sub>·NaOH per unit cell. We believe this to be the first reported example of an anhydrous alkali metal phosphate hydroxide, although  $\text{Na}_2\text{Zn}(\text{PO}_4)(\text{OH})$  is known,<sup>18</sup> as are a great number of phases<sup>19</sup> of the types  $(M<sup>H</sup>)<sub>5</sub>(PO<sub>4</sub>)<sub>3</sub>(OH)$  and  $(M<sup>H</sup>)<sub>2</sub>(PO<sub>4</sub>)(OH).$ Our attempts to prepare the analogous salts  $Na<sub>3</sub>PO<sub>4</sub>·NaF$  and Na3PO4.NaC1 according to *eq* 3 were unsuccessful, consistent with phase diagram studies<sup>20,21</sup> in the systems  $Na_3PO_4-NaF$ and  $Na<sub>3</sub>PO<sub>4</sub>–NaCl.$ 

$$
Na2HPO4 + 1/2Na2CO3 + NaX \xrightarrow[20 h]{600 °C}
$$
  
\n
$$
Na3PO4NaX + 1/2H2O + 1/2CO2
$$
 (3)

**Phase** II. The strong lines in the diffraction patterns of IIA and IIB (Table 111) are grouped about the *d* spacings reported<sup>10-12</sup> for the high-temperature, cubic form of  $Na_3PO_4$ , suggesting a structural relationship between phase I1 and cubic  $Na<sub>3</sub>PO<sub>4</sub>$ . These diffraction patterns also bear a close resemblance to that reported by Kizilyalli and Welch<sup>13</sup> for "pseudo-cubic  $\gamma$ -Na<sub>3</sub>PO<sub>4</sub>". The strong reflections in the 22 "C diffraction pattern of IIA have been indexed on the basis of a triclinic unit cell with dimensions  $a = 7.522$  (3)  $\AA$ ,  $b =$ 7.378 (3) Å,  $c = 7.341$  (3) Å,  $\alpha = 91^{\circ}$  46 (3)',  $\beta = 90^{\circ}$  49 (3)',  $\gamma = 89^{\circ}$  4 (3)', similar to the unit cell of cubic Na<sub>3</sub>PO<sub>4</sub> in which  $a = 7.410 \text{ Å}$  (Table III). The diffraction pattern also contained a large number of weak lines, many of which could not be indexed on the basis of this unit cell. Some of these lines could be assigned to traces of other phases, but some of them probably arose from a superlattice structure. The diffraction pattern of IIB was somewhat diffuse, but it was tentatively indexed on the basis of a second triclinic unit cell, with  $a = 7.505$  (4)  $\text{\AA}$ ,  $b = 7.467$  (4)  $\text{\AA}$ ,  $c = 7.292$  (4)  $\text{\AA}$ ,  $\alpha$ = 90° 34 (4)',  $\beta$  = 91° 10 (4)',  $\gamma$  = 90° 6 (4)' (Table IV). The four lines attributed to IIC correspond to a cubic unit cell with  $a = 7.461$  Å (Table IV).

As noted above, the phases corresponding to 11, isolated by Broadbent et al.,<sup>1</sup> gave an average analysis close to the formulation  $11Na_2O·4P_2O_5·3H_2O$ , or  $Na_{2,75}H_{0,25}PO_4·0.25H_2O$ . Thermogravimetric analysis of our phase IIA, obtained from a solution of starting composition  $Na/P = 2.83$ , showed a weight loss above 250  $\degree$ C of 2.8%, consistent with the total dehydration of  $\text{Na}_{2.83}\text{H}_{0.17}\text{PO}_4.0.17\text{H}_2\text{O}$  according to eq 4.<br>  $\text{Na}_{2.83}\text{H}_{0.17}\text{PO}_4.0.17\text{H}_2\text{O} \rightarrow \text{Na}_{2.83}\text{H}_{0.17}\text{PO}_4.0.17\text{H}_2\text{O} \rightarrow \text{Ca}_{2.83}\text{N}_{0.17}\text{PO}_4.0.17\text{N}_{0.17}\text{O} \rightarrow \text{Ca}_{2.83}\text{N$ 

$$
0.255H_2O + 0.085Na_4P_2O_7 + 0.83Na_3PO_4
$$
 (4)

The products of this analysis, which was terminated at 1000 "C, were confirmed by X-ray diffraction to be a mixture of  $Na_4P_2O_7$  (minor) and  $\alpha$ -Na<sub>3</sub>PO<sub>4</sub> (major). It is therefore likely that the type I1 phases can be described by the general formula  $Na_{3-x}H_xPO_4 \cdot xH_2O$ . This could be envisaged as  $[Na_{3-x}]$  $(H_3O)_xPO_4$ ,  $[Na_{3-x}(HPO_4)_x(PO_4)_{1-x}xH_2O]$ , or  $[Na_{3-x}$  $(HPO<sub>4</sub>)<sub>2x</sub>(PO<sub>4</sub>)<sub>1-2x</sub>(OH)<sub>x</sub>$ , or a dynamic equilibrium or intermediate composition between these forms. In view of the strong basicity of  $PO_4^{3-}$ , the first form is unlikely to contribute significantly to the true formulation. We suggest that the structure of phase I1 is formally derived from that of cubic  $Na<sub>3</sub>PO<sub>4</sub>$  by the protonation of *x* of the phosphate ions per formula unit and the substitution of  $x$  sodium ions by water molecules, perhaps with some further redistribution of protons between water and phosphate moieties. The distortion of the structures of IIA and IIB from cubic symmetry and the possible existence of superlattice structures are probably due to hydrogen bonding interactions and/or ordering of sodium substitution sites.

The diffraction pattern of IIA was monitored as a function of temperature between 25 and 285 °C. With increasing temperature, the diffraction lines progressively shifted and coalesced, such that the 285 °C pattern can be indexed on the basis of a rhombohedral unit cell:  $a = 7.482$  (2) Å,  $\alpha = 89^{\circ}$ **13** (2)' (Table 111). Slight broadening of some lines suggests a very slight distortion from true rhombohedral symmetry at this temperature, the upper limit of the Guinier-Simon system. Extrapolating the progression of the diffraction pattern, one observes that it is likely that the structure achieves true rhombohedral symmetry close to 300 °C. The rhombohedral cell is again closely related to the cubic  $Na<sub>3</sub>PO<sub>4</sub>$  unit cell. The continuously monitored progress of the diffraction pattern with increasing temperature showed that  $(11\bar{1})$  of the triclinic cell becomes (11<sup>T</sup>) of the rhombohedral cell. We attribute the increase in symmetry to the disappearance of the orienting influence of hydrogen bonding; the process is reversible. We have observed a similar progressive change in the diffraction pattern of  $Na<sub>2</sub>HPO<sub>4</sub>$ , corresponding to transformation from a monoclinic low-temperature unit cell to an essentially isodimensional orthorhombic cell at 285 °C.

In a similar experiment with IIB, the structure remained triclinic at 285  $\rm ^o\rm C$  (Table IV). The reason for this is obscure, but a different mode of modification of the cubic  $Na<sub>3</sub>PO<sub>4</sub>$ structure may be involved. It is possible that IIB is a dehydrated phase,  $Na_{3-x}H_xPO_4\cdot yH_2O$ , with  $y < x$ . Unfortunately, no further discussion of the structures of type I1 phases is justified in the absence of single-crystal crystallographic structural data. Preliminary single-crystal studies of IIA have confirmed that the unit cell is related to that of cubic  $Na<sub>3</sub>PO<sub>4</sub>$ , but extensive twinning has so far prevented the acquisition of accurate structural data.

**Phase 111.** Although the diffraction pattern of I11 bears some similarity to the phase I1 patterns, we have not been able to index it on this basis. On heating III dry to  $285 \degree C$ , we observed that the diffraction pattern exhibits no changes other than those attributable to thermal expansion and partial decomposition.

#### **Phase Diagram**

The high-temperature equilibrium solid phases for the system  $Na<sub>2</sub>O-P<sub>2</sub>O<sub>5</sub>-H<sub>2</sub>O$  have been deduced from Table I, the structural information in the preceding section, and literature solubility data.<sup>1-3</sup> The results of these deductions are illustrated in Figure 2, which is a schematic representation of Broadbent's phase diagram,' modified to account for the results of this study. To optimize the use of space in the triangular diagram, we chose the three components to be  $Na<sub>2</sub>O, P<sub>2</sub>O<sub>5</sub>·1.5Na<sub>2</sub>O, and a 100-mod quantity of H<sub>2</sub>O. The$ solubility isotherm in Figure 2 approximates to the literature data,<sup>1-3</sup> except for Na/P > 9, where the mole fraction of Na<sub>2</sub>O was reduced for clarity because of the extremely high solubility of NaOH at 300 °C. The solubility studies<sup>1-3</sup> suggest that the isothermal phase diagram in Figure 2 is qualitatively correct from about 275 to 325 °C, except that the stability of NaH2P04.Na2HP04 above 300 **OC** is unconfirmed. The detailed arguments presented below are based on Findlay's discussion<sup>22</sup> of the isothermal evaporation of ternary systems



**Figure 2.** Schematic representation of part of the  $Na<sub>2</sub>O-P<sub>2</sub>O<sub>5</sub>-H<sub>2</sub>O$ isothermal phase diagram near 300 °C. The water mole fractions for Na/P > 9.0 were distorted for clarity. The phase  $Na<sub>3</sub>PO<sub>4</sub>NaOH$ may exist as an incongruently saturating equilibrium phase in the presence of liquid water when  $Na/P > 9.0$ , as shown by the dashed line.

and proceed from right to left along the ternary diagram in Figure 2.

The solid phase in equilibrium with saturated aqueous NaOH at 290  $\degree$ C is known to be  $\beta$ -NaOH.<sup>23,24</sup> Presumably the NaOH $\cdot$ H<sub>2</sub>O observed at Na/P = 9.3 formed from traces of water in the apparatus on cooling. Phase I,  $Na<sub>3</sub>PO<sub>4</sub>$  NaOH, may be an equilibrium phase in the presence of the saturated aqueous solutions, or it may have been formed by a solid-state reaction between NaOH and phase I1 while the mixture was cooling down in the dry autoclave. The analogous reaction between NaOH and Na<sub>3</sub>PO<sub>4</sub> solids yielded Na<sub>3</sub>PO<sub>4</sub>. NaOH rapidly above 90 °C. Broadbent's solubility data<sup>1</sup> indicate that between NaOH and N<sub>a3</sub>PO<sub>4</sub> solids yielded Na<sub>3</sub>PO<sub>4</sub>. NaOH<br>hetween NaOH and Na<sub>3</sub>PO<sub>4</sub> solids yielded Na<sub>3</sub>PO<sub>4</sub>. NaOH<br>I does not coexist with solution compositions Na/P ≤ 9. It<br>might, however, coexist with sincongruentl might, however, coexist with incongruently saturated solutions<sup>22</sup> at still higher Na/P values. If so, the invariant points a and b (Figure 2) corresponding to the  $I/\beta$ -NaOH and II/I equilibria would exist at solution compositions with Na/P > 9. Otherwise,  $Na_{3-x}H_xPO_4 \cdot xH_2O$  and  $\beta$ -NaOH would form an invariant solution of composition c, as shown by the dashed lines.

Solubility studies $1-3$  clearly show that variants of phase II are the equilibrium solids associated with saturated solutions with Na/P values from about 2.15 to at least 9.0. The data in Table I confirm that congruent precipitation occurs at Na/P =  $2.80 \pm 0.05$ , point d in Figure 2. Our recovery of phases other than I1 from solutions with starting compositions within the range  $2.15 \leq \text{Na}/\text{P} \leq 9$  is due to changes in the solution composition arising from incongruent precipitation during evaporation.<sup>22</sup> The  $\alpha$ -Na<sub>3</sub>PO<sub>4</sub> recovered at Na/P = 3.15 and 3.00 undoubtedly formed from variants of phase I1 close to the stoichiometric composition,  $\gamma$ -Na<sub>3</sub>PO<sub>4</sub>, at 300 °C. Mair<sup>10</sup> reported that solid solutions of  $Na<sub>2</sub>CO<sub>3</sub>$  or  $Na<sub>2</sub>SO<sub>4</sub>$  in  $Na<sub>3</sub>PO<sub>4</sub>$ revert to the  $\alpha$ -Na<sub>3</sub>PO<sub>4</sub> structure on cooling only at solute concentrations below about 4 mol %. The solubility data of both Ravich and Shcherbakova<sup>3</sup> and Broadbent et al.<sup>1</sup> indicate that the equilibrium solid composition tends toward  $Na<sub>3</sub>PO<sub>4</sub>$ as Na/P in solution is increased. Significant amounts of solid products with  $Na/P > 2.95$  would thus have been precipitated in the experiments in which we observed  $\alpha$ -Na<sub>3</sub>PO<sub>4</sub>.

Heating experiments in the Guinier-Simon X-ray camera did not reveal formation of III from IIA and  $Na<sub>2</sub>HPO<sub>4</sub>$ , dry,



**Figure 3.** Vapor pressure data for condensation reactons of NaH<sub>2</sub>PO<sub>4</sub>,  $Na<sub>2</sub>HPO<sub>4</sub>$ , and  $Na<sub>2</sub>H<sub>2</sub>P<sub>2</sub>O<sub>7</sub>$  and for vapor-saturated pure water. Experimental data for  $\text{NaH}_2\text{PO}_4$  and  $\text{Na}_2\text{H}_2\text{P}_2\text{O}_7$  are from ref 33 (O, *0)* and ref 34 *(0, 0).* Vertical bars refer to the standard deviations in the calculated vapor pressure curves.

at 285 °C. It is therefore unlikely that III formed by solid-state reactions subsequent to precipitation, and we conclude that it is an equilibrium phase. The solubility data<sup>1,2</sup> suggest that III forms an invariant solution with  $Na<sub>2</sub>HPO<sub>4</sub>$  or  $Na<sub>4</sub>P<sub>2</sub>O<sub>7</sub>$ at  $Na/P \approx 2.15$ , composition h in Figure 2, and that no congruently saturated solutions exist between point d and point h. It is likely that the incongruently saturating invariant solution<sup>22</sup> in equilibrium with II and III is also close to  $Na/P$  $= 2.15$ , as shown by point g in the phase diagram. Otherwise, I11 would have been observed by other workers. The results *of* our study neither confirm nor contradict Broadbent's observation' of liquid-phase separation in the composition range  $2.0 \leq \text{Na}/\text{P} \leq 2.15$ . The solubility data are not sufficiently precise to show which solid phase is in equilibrium with the two liquid phases at invariant compositions e and f. Figure 2 was drawn on the assumption that it is phase 111, but it could equally well be phase I1 with the II/III invariant composition, g, occurring between e and d.

The major phase observed at  $Na/P = 2.00$  was  $Na<sub>2</sub>HPO<sub>4</sub>$ with traces of  $Na_4P_2O_7$  which presumably formed at 290 °C via the condensation reaction *(5)* when the water vapor

$$
2Na_2HPO_4 \rightleftharpoons Na_4P_2O_7 + H_2O \tag{5}
$$

pressure was lowered below the saturation value. Broadbent et al.<sup>1</sup> demonstrated that  $Na_4P_2O_7$  is the stable phase at 300  $\degree$ C. To ensure that the Na<sub>2</sub>HPO<sub>4</sub> which we observed was not formed from  $Na_4P_2O_7$  by reaction with water vapor during cooling, we prepared  $\text{Na}_4\text{P}_2\text{O}_7$  by evaporation at 330 °C (Table I) and then exposed it to 7 MPa of water vapor at 290 <sup>o</sup>C and cooled in the usual way. The small amount of  $Na<sub>2</sub>HPO<sub>4</sub>$  (<20%) which formed during this and a similar run using finely ground anhydrous commercial  $Na<sub>4</sub>P<sub>2</sub>O<sub>7</sub>$  led us to conclude that the bulk of the  $Na<sub>2</sub>HPO<sub>4</sub>$  recovered from solutions evaporated at 290 °C could not have formed by reaction of  $\text{Na}_4\text{P}_2\text{O}_7$  with the vapor. Hence,  $\text{Na}_2\text{HPO}_4$  is the equilibrium phase at 290 °C, and the transition from  $Na<sub>2</sub>HPO<sub>4</sub>$  to  $Na<sub>4</sub>P<sub>2</sub>O<sub>7</sub>$  as the saturating phase occurs between 290 and 300 **"C.** The water vapor pressures corresponding to the  $\text{Na}_4\text{P}_2\text{O}_7/\text{Na}_2\text{HPO}_4$  equilibrium were calculated from literature thermodynamic data<sup>24-30</sup> (see supplementary Appendix) and are plotted in Figure **3.31932** The calculated vapor pressure curve intersects the water vapor pressure line<sup>30</sup> at 320

 $\pm$  25 °C, in good agreement with our observation. The reliability of the vapor pressure calculations is confirmed by the agreement with experimental values<sup>33,34</sup> for the NaH<sub>2</sub>PO<sub>4</sub>/  $Na<sub>2</sub>H<sub>2</sub>P<sub>2</sub>O<sub>7</sub>$  and the  $Na<sub>2</sub>H<sub>2</sub>P<sub>2</sub>O<sub>7</sub>/(NaPO<sub>3</sub>)<sub>x</sub>$  equilibria at temperatures where the reactions are reversible  $(>145 \degree C \text{ and } > 245 \degree C$ , respectively). The lack of other products at Na/P  $=$  2.00 in our experiments is consistent with other evidence of congruent precipitation from solution<sup>1,7,8</sup> at this composition. $3$ 

The major crystalline phase recovered at  $Na/P = 1.6$  was  $Na<sub>2</sub>HPO<sub>4</sub>NaH<sub>2</sub>PO<sub>4</sub>$ . Since it decomposed to  $Na<sub>4</sub>P<sub>2</sub>O<sub>7</sub>$  and an amorphous orthophosphate during recovery at somewhat reduced water vapor pressures, we conclude that it is an equilibrium precipitate whose upper limit of stability is not much higher than 290 °C. It probably precipitates congruently at  $Na/P = 1.5$ , but autoclave corrosion problems prohibited experimentation at this composition. The unidentified phases observed at Na/P = 1.6 and **1.8** may well have been metastable phases or products of partial decomposition. If so, there is an invariant composition, j, at which  $Na<sub>2</sub>HPO<sub>4</sub>$  and  $Na<sub>2</sub>HPO<sub>4</sub>·NaH<sub>2</sub>PO<sub>4</sub>$  coexist. If any of the unidentified solids are equilibrium phases, further invariant points must exist.

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**Registry No. Na<sub>3</sub>PO<sub>4</sub>.NaOH**, 70850-23-6; Na<sub>3</sub>PO<sub>4</sub>, 7601-54-9;  $Na_{2,5}H_{0,5}PO_4$ , 70659-31-3; NaOH, 1310-73-2; Na<sub>2</sub>HPO<sub>4</sub>.NaH<sub>2</sub>PO<sub>4</sub>, 65185-91-3; Na<sub>2</sub>HPO<sub>4</sub>, 7558-79-4; Na<sub>4</sub>P<sub>2</sub>O<sub>7</sub>, 7722-88-5; NaH<sub>2</sub>PO<sub>4</sub>, 7558-80-7;  $Na<sub>2</sub>H<sub>2</sub>P<sub>2</sub>O<sub>7</sub>$ , 7758-16-9;  $Na<sub>3</sub>P<sub>3</sub>O<sub>9</sub>$ , 7785-84-4;  $Na<sub>2</sub>O<sub>9</sub>$ 1313-59-3;  $P_2O_5$ , 1314-56-3;  $H_2O$ , 7732-18-5.

**Supplementary Material Available:** Appendix, partial pressure data calculation; Table V, X-ray diffraction data for phase 111; Table VI, X-ray diffraction data for the metastable polymorph of  $Na<sub>2</sub>HP Q_4$ -NaH<sub>2</sub>PO<sub>4</sub>; Table VII, thermodynamic data for selected Na<sub>2</sub>O-H<sub>2</sub>O solids (4 pages). Ordering information **is** given on any current masthead page.

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- (35) E. A. M. Wetton (personal communication) has pointed out that both  $Na_4P_2O_7$  and  $Na_2HPO_4$  must coexist as equilibrium phases in the region hij, over a finite but probably very small temperature range, since the activity of water is not constant throughout **this** region. Invariant solution compositions corresponding to the  $\text{Na}_4\text{P}_2\text{O}_7/\text{Na}_2\text{HPO}_4$  equilibrium would exist between i and j and/or between i and h in this temperature range.

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# **Kinetics and Thermodynamics of Dissociation of Nickel( 11)-Isoquinoline Complexes in Water, Ethanol, and 1-Propanol**

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Rate constants and associated activation parameters for the dissociation of  $(solv)_5Ni(isoquin)^{2+} (solv = solvent, isoquin)$ = isoquinoline) in ethanol, 1-propanol, and water and of (solv)<sub>S</sub>NiSCN<sup>+</sup> in 1-propanol have been measured kinetically by stopped-flow spectrophotometry using  $H_3O^+$  as the decomposing ion for (solv)<sub>s</sub>Ni(isoquin)<sup>2+</sup> and copper(II) for (solv)<sub>s</sub>NiSCN<sup>+</sup>. These data are compared with previously reported results on dissociation of (solv)<sub>s</sub>Ni(isoquin)<sup>2+</sup> and (solv)<sub>s</sub>NiSCN<sup>+</sup> in several donor solvents. Pathways for dissociation of both (solv)<sub>s</sub>Ni(isoquin)<sup>2+</sup> and (solv)<sub>s</sub>NiSCN<sup>+</sup> in the solvents studied are accommodated within an  $I_d$ -type mechanism.

#### **Introduction**

An improved understanding of solvent properties can be obtained from studies of reactions in which solvent molecules are direct participants, such as in the solvolysis of ligandsubstituted complexes. It has been shown<sup>1,2</sup> that measurements of dissociation rates of  $(solv)_5Ni(isoquin)^{2+}$  (solv = solvent, isoquin = isoquinoline) in water, methanol, acetonitrile, and dimethyl sulfoxide and of  $(solv)_{5}NiSCN^{+}$  in water, methanol, acetonitrile, dimethyl sulfoxide, and dimethylformamide can be interpreted through an  $I<sub>d</sub>$ -type mechanism represented by *eq* 1, where solv is a solvent molecule and L is either iso- $LNi(solv)_5 + solv \rightleftharpoons solv,(solv)_5Ni-L \rightleftharpoons solv-Ni(solv)_5,L$ 

**(1)** 

quinoline or thiocyanate ion (charges omitted). It has also been shown that dissociation rate constants at a single temperature do not bear a systematic relation to the donor properties of the solvent, but enthalpies of activation for dissociation of both (solv)<sub>5</sub>Ni(isoquin)<sup>2+</sup> and (solv)<sub>5</sub>NiSCN<sup>+</sup> bear an inverse linear relation to solvent donor properties as expressed by the Gutmann donor number. $3$ 

We report in this work studies of the dissociation of  $(solv)_{5}Ni(isoquin)^{2+}$  in water, ethanol, and 1-propanol and of  $(\text{solv})$ . NiSCN<sup>+</sup> in 1-propanol over a range of temperatures. Enthalpies and entropies of activation for dissociation in these solvents have been calculated; the results, in conjunction with previous work on these systems in other solvents, $^{1,2}$  are discussed in terms of mechanisms and solvent donor properties.

#### **Experimental Section**

Solvents. Commercial 1-propanol (Baker Analyzed, assay 99%) was refluxed for 12 h over BaO and fractionally distilled in the absence of air. The middle fraction, collected over alumina previously baked 24 h at 550 °C, was shaken for 12 h and then fractionally distilled. The middle cut from the second distillation was fractionally distilled twice more from metallic sodium in the absence of air. The purified solvent was used within 24 h of the final distillation. Pure ethanol, USP (US. Industrial Chemicals), was used as obtained without further treatment. Singly distilled water was redistilled from alkaline KMn0, before use.

**Reagents.** Nickel(I1) was introduced as the perchlorate salt. In all solvents, nickel(I1) perchlorate monohydrate, prepared **as** previously described,<sup>4</sup> was used. Copper(II) was introduced as hexakis(acetonitrile)copper(II) perchlorate and was prepared **as** before? Perchloric acid (Mallinckrodt, analytical reagent, **70%)** was used as received. described before.<sup>4</sup> Potassium thiocyanate (J. T. Baker, Baker Analyzed, assay 100%) was dried at 60 °C under vacuum for 3 days.<br>Instruments and Experimental Procedure. All nonaqueous solutions

were prepared and dispensed in a drybox (Kewaunee Scientific) under dry nitrogen by using glassware that had been oven-dried at  $100^{\circ}$ C for several days. All standard procedures for handling nonaqueous solutions were followed. The first-order dissociation rate constant of  $(solv)_{5}Ni(isoquin)^{2+}$  in perchloric acid was determined by measuring the rate of formation of isoquinH<sup>+</sup> by following the rate of change of transmittance upon mixing of a solution of  $(solv)_{5}Ni(isoquin)^{2+}$ with  $HClO<sub>4</sub>$  at a wavelength where isoquin $H<sup>+</sup>$  absorbs strongly and  $(solv)_5Ni(isoquin)^{2+}$  and  $HClO<sub>4</sub>$  absorb but little. A stopped-flow spectrophotometer (Durrum Instrument Co., Model D110) was used for all kinetic measurements. A description of the instrument along with modifications for better temperature control has been given.<sup>6</sup> Solutions of (solv)<sub>s</sub>Ni(isoquin)<sup>2+</sup> were prepared in dry volumetric flasks by adding measured volumes of a standard nickel(I1) solution to a known volume of a stock isoquinoline solution and diluting to volume with solvent. In all solutions, the concentration of nickel(I1) was in **50-** to 100-fold excess of the isoquinoline concentration to ensure the formation of the mono complex only. The total concentration of perchloric acid in the final mixed solution was held higher than the total concentration of nickel(I1) by a ratio of 2 to 1 in water and 1-propanol and 6 to 1 in ethanol. In 1-propanol, kinetic measurements were done in both the presence and absence of acetic anhydride, the concentration of acetic anhydride being held equal to the concentration of perchloric acid or held at double the concentration of perchloric acid. Absorbance measurements were made at 332 nm in water and 1-propanol and at 334 nm in ethanol. Dissociation rate constants of

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